Spatial resolution and switching field of magnetic force microscope tip coated with FePd-alloy thin film

Shinji Ishihara, Mitsuru Ohtake, and Masaaki Futamoto

Faculty of Science and Engineering, Chuo University, 1-13-27 Kasuga, Bunkyo-ku, Tokyo, 112-8551, Japan

Abstract. Magnetic force microscope (MFM) tips are prepared by coating Si tips of 4 nm radius with $L1_0$ ordered FePd-alloy films varying the thickness in a range between 10 and 80 nm. The effects of coating thickness on spatial resolution and switching field of MFM tip are investigated. As the thickness increases from 10 to 20 nm, the MFM signal detection sensitivity is improved and the resolution improves from 12.7 to 7.9 nm. With further increasing the thickness, the resolution decreases due to increase of tip radius. Magnetic bits of 15.9 nm length of a perpendicular medium recorded at 1600 kilo-flux-change-per-inch are distinguishable in the MFM image observed by using a tip coated with 20-nm-thick FePd film. The switching field monotonically increases from 0.70 to 1.50 kOe with increasing the coating thickness from 10 to 80 nm. The present study has shown that it is possible to prepare an MFM tip with spatial resolution better than 10 nm and switching field higher than 1 kOe by coating a sharp Si tip with an $L1_0$ ordered FePd-alloy film.

1 Introduction

Magnetic force microscopy (MFM) has been widely used to investigate the magnetization structures of magnetic devices like hard-disk-drive (HDD) media, permanent magnets, etc. MFM tips are generally prepared by coating non-magnetic sharp tips with magnetic materials [1]–[7]. The tip shape and the magnetic property of coated material influence the spatial resolution and the switching field ($H_{sw}$) of MFM tip.

The areal density of HDD medium is now approaching to 1 Tb/in$^2$ and expected to further increase. Future HDD media are considered to consist of magnetic materials with very high uniaxial magnetocrystalline anisotropy energies such as FePt and SmCo$_5$. Therefore, realization of high $H_{sw}$ is required in addition to high-resolution better than 10 nm for MFM observations of such ultra-high-density media. MFM tips with high $H_{sw}$ have been prepared by annealing tips coated with $L1_0$ ordered FePt- and CoPt-alloy films followed by focused-ion-beam milling [6, 7]. However, the resolution of commercially available MFM tip coated with $L1_0$ ordered film is limited at around 20 nm.

In order to improve the resolution, it is necessary to reduce the magnetic volume interacting with a magnetic sample. In our previous studies [8]–[10], MFM tips were prepared by coating Si tips with magnetic materials such as Co-Cr-Pt, Co, Fe, Ni, Ni-Fe, and Fe-B. Spatial resolutions around 10 nm were obtained by adjusting the coating film thickness. An FePd-alloy film is reported to show a higher degree of $L1_0$ ordering than FePt- and CoPt-alloy films, when prepared by using a similar high-temperature process [11]. In the present study, MFM tips are prepared by coating Si tips with FePd films. The effects of coating thickness on resolution and $H_{sw}$ are investigated.

2 Experimental procedure

MFM tips were prepared by coating base-Si tips of 4 nm radius with thin films by employing a radio-frequency (RF) magnetron sputtering system with base pressures lower than $4 \times 10^{-7}$ Pa. Fe$_{50}$Pd$_{50}$ (at. %) and MgO targets of 3 in diameter were used. The distance between target and Si tip was 150 mm. The Ar gas pressure was kept constant at 0.67 Pa. The RF powers for FePd and MgO targets were fixed at 35 and 200 W, where the deposition rates were 0.020 and 0.015 nm/s, respectively. MgO underlayers and FePd films were sequentially formed on base-Si tips at room temperature (RT). The MgO underlayer was introduced to avoid atomic mixing between the Si tip and the FePd film. The thickness of MgO underlayer was fixed at 5 nm, while that of FePd film was varied in a range between 10 and 80 nm. The thicknesses were estimated for films deposited on flat substrates which were located near the base-Si tips. After film formation, the tips were annealed at 600 °C for 1 h in the ultra-high vacuum chamber to obtain ordered $L1_0$ structure.

The tip shapes were observed by scanning electron microscopy (SEM). In order to investigate the structural...
medium with the areal density of 163 Gb/in² were used as
inch (kFCI) and a commercial HDD perpendicular
linear densities from 1000 to 1700 kilo-flux-change-per-
magnetic pole. A perpendicular medium recorded at
along the tip axis so that the tip top possesses the south
pressures lower than 0.1 Pa. MFM tips were magnetized
magnetic properties were measured by using a vibrating

$$R_a = \left( \frac{\text{area}}{\pi} \right)^{1/2}$$

structure was investigated by
surface morphology was observed by AFM. The crystal
observation samples. The quality factor value, the
radius.

Figures 1(a)–(d) show the AFM images of a Si substrate and of FePd/MgO bi-layer films formed on Si substrates. The FePd-layer thicknesses are (a) 20, (b) 40, (c) 60, and (d) 80 nm. (e, f) Dependences of FePd-layer thickness on (e) $R_a$ and (f) island radius.

and the magnetic properties of coated films, FePd/MgO bi-layer films were also deposited on flat Si substrates with natural oxide layers at RT and the samples were then annealed at 600 °C for 1 h under conditions similar to the case of MFM tip preparation. The arithmetical mean surface roughness ($R_a$) value of a Si substrate measured by atomic force microscopy (AFM) was 0.2 nm. The film surface morphology was observed by AFM. The crystal structure was investigated by $\theta$-2θ scan X-ray diffraction (XRD) with Cu-Kα radiation ($\lambda = 0.15418$ nm). The magnetic properties were measured by using a vibrating sample magnetometer.

MFM observation was carried out at RT under pressures lower than 0.1 Pa. MFM tips were magnetized along the tip axis so that the tip top possesses the south magnetic pole. A perpendicular medium recorded at linear densities from 1000 to 1700 kilo-flux-change-per-inch (kFCI) and a commercial HDD perpendicular medium with the areal density of 163 Gb/in² were used as observation samples. The quality factor value, the distance between tip and observation sample, and the scanning speed were respectively 3000–6000 (dimensionless quantity), 5–10 nm, and 1.4 µm/s.

3 Results and discussion

Figures 1(a) and (b)–(d) show the AFM images of a Si substrate and of FePd/MgO bi-layer films with different FePd-layer thicknesses formed on Si substrates, respectively. Figures 1(e) and (f) show the thickness dependences on $R_a$ and island radius, respectively. Here, the island radius is estimated by using the relation,

$$(\text{radius}) = \left( \frac{\text{area}}{\pi} \right)^{1/2}$$

With increasing the thickness, the $R_a$ and the island radius increase, suggesting that the radius of MFM tip increases. Figures 2(a-1)–(c-1) show the SEM images observed for a base-Si tip and of MFM tips coated with FePd(20 nm)/MgO(5 nm) and FePd(40 nm)/MgO(5 nm) films. Island-like growth of FePd/MgO bi-layer films on Si tips are recognized, similar to the case of film growth on flat Si substrates. Figures 2(a-2)–(c-2) show the high-magnification views of top parts of the tips shown in figures 2(a-1)–(c-1), respectively. The tip radius increases with increasing the thickness, as expected from the AFM observations.

Figures 3 shows the XRD spectra of FePd/MgO bi-layer films formed on Si substrates. The FePd-layer thicknesses are (a) 20, (b) 40, (c) 60, and (d) 80 nm. The intensity is shown in a logarithmic scale.

Figure 3 shows the XRD spectra of FePd/MgO bi-layer films formed on Si substrates. FePd(001) superlattice reflections are recognized in addition to FePd(002), (022), and (111) fundamental reflections for all the samples. FePd(110) superlattice and FePd(220) and (022) fundamental reflections are also observed when the FePd-layer thickness increases beyond 60 nm. This is because the volumes of FePd(220) and (022) crystals are large enough to be detected by XRD for the FePd-layer thicknesses of 60 and 80 nm. The order degree is almost kept constant at around 0.8 for the FePd-layer thickness. $L1_0$ ordered phase is also expected to be formed in the FePd/MgO bi-layer films prepared on Si tips.

Figures 4(a)–(d) show the magnetization curves of FePd/MgO bi-layer films formed on Si substrates. Figure 4(c) shows the thickness dependences on in-plane and
out-of-plane coercivities \( (H_c) \). As the thickness increases, the in-plane and the out-of-plane \( H_c \) increase. The result indicates that the \( H_{sw} \) of FePd-coated MFM tip increases with increasing the FePd-layer thickness.

Figures 5(a-1)–(g-1) show the MFM images of a same area of perpendicular medium recorded at 1000 kFCI (bit length: 25.4 nm) observed by using MFM tips coated with FePd/MgO bi-layer films with different FePd-layer thicknesses. Figures 5(a-2)–(g-2) show the MFM signal profiles along the dotted lines in figures 5(a-1)–(g-1), respectively. Magnetic bit images corresponding to 1000 kFCI are clearly recognized. Thus, the MFM resolution is between 15.9/2 = 7.9 nm (1600 kFCI) and 14.9/2 = 7.5 nm (1700 kFCI), that is, 7.7 ± 0.2 nm. Figure 7 summarizes the resolutions of tips with different FePd-layer thicknesses. As the thickness increases from 10 to 20 nm, the resolution improves from 12.1 to 7.7 nm, due to that the signal detection sensitivity is improved. With further increasing the thickness, the resolution gradually decreases. This is because the tip radius increases with increasing the thickness. The MFM resolution is apparently influenced by both the detection sensitivity and the tip radius.

For \( H_{sw} \) estimation, MFM observations of an HDD medium (163 Gb/in\(^2\)) were performed for MFM tip repeatedly before and after applying a magnetic field. The magnetic field direction was opposite to the magnetization direction of MFM tip and the field was increased in a step-wise of 0.05 kOe. The \( H_{sw} \) was estimated as the magnetic field where the contrast of MFM image was reversed. Figures 8(a)–(c) show the MFM images of a same area of HDD medium observed by using a tip coated with 80-nm-thick FePd film before and after applying magnetic fields. When a field of 1.45 kOe is applied for the tip [figure 8(b)], the MFM contrast does not change, when compared with the initial magnetization state shown in figure 8(a). With increasing the field up to 1.50 kOe, the contrast is reversed, as shown in figure 8(c). Thus, the \( H_{sw} \) is between 1.45 and 1.50 kOe (i.e. 1.475 ± 0.025 kOe). Figure 8(d) shows the dependence of FePd-layer thickness on \( H_{sw} \). With increasing the thickness from 10 to 80 nm, the \( H_{sw} \) increases from 0.675 to 1.475 kOe. The \( H_{sw} \) of MFM tip coated with FePd/MgO bi-layer film is larger than the \( H_c \) of FePd/MgO bi-layer film deposited on flat Si substrate for respective thickness. The enhancement is possibly due to an influence of shape magnetic anisotropy. The magnetization state of MFM tip has been studied by using micromagnetic simulations [12–14]. It seems necessary to investigate the magnetization reverse process of FePd-coated tip by such theoretical calculations. The present study apparently shows that it is possible to prepare an MFM tip with resolution better than 10 nm and \( H_{sw} \) higher than 1 kOe by coating a sharp Si tip with an \( L1_0 \) ordered FePd film.
Fig. 6. (a-1)–(a-6) MFM images of a perpendicular medium recorded at (a-1) 1000, (a-2) 1200, (a-3) 1400, (a-4) 1500, (a-5) 1600, and (a-6) 1700 kFCI observed by using an MFM tip coated with 20-nm-thick FePd film. (b-1)–(b-6) Enlarged views of the areas surrounded by white square lines in (a-1)–(a-6), respectively. (c-1)–(c-6) Signal profiles along the dotted lines in (a-1)–(a-6), respectively. (d-1)–(d-6) Power spectra analyzed for the magnetic bit images of (a-1)–(a-6), respectively.

Fig. 7. Dependence of FePd-layer thickness on resolution.

4 Conclusions

MFM tips are prepared by coating Si tips with \( L_{10} \) ordered FePd-alloy films. The effects of coating thickness on spatial resolution and switching field are investigated. With increasing the thickness from 10 to 20 nm, the resolution improves from 12.7 to 7.9 nm, due to improvement of signal detection sensitivity. As the thickness further increases, the resolution decreases. The reason is due to that the tip radius increases with increasing the thickness. The MFM resolution is affected by both the signal detection sensitivity and the tip radius. The switching field monotonically increases from 0.70 to 1.50 kOe with increasing the coating thickness from 10 to 80 nm. An FePd-coated MFM tip is useful to observe the magnetization structures of future high-density HDD media.

Acknowledgments

Authors thank Ms. Noriko Saidoh of RIMCOF for SEM observations. A part of this work was supported by METI and JSPS (Grant-in-Aid for Scientific Research, No. (C) 22560302), Japan.

References