

## Comparative study of Ho:Y<sub>3</sub>Al<sub>5</sub>O<sub>12</sub> and Ho:Y<sub>2</sub>O<sub>3</sub> transparent ceramics synthesized from laser ablated nanopowders

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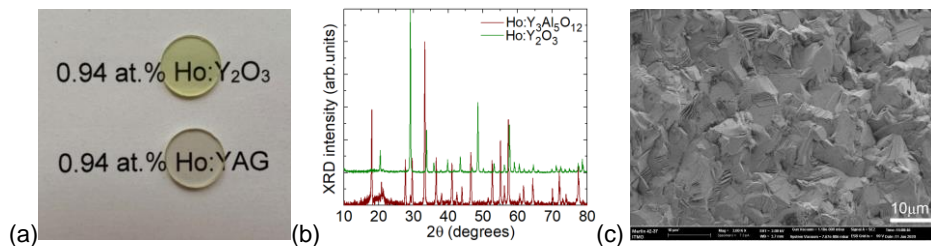
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Holmium (Ho<sup>3+</sup>) ions are known for their emission at 2 μm originating from the <sup>5</sup>I<sub>7</sub> → <sup>5</sup>I<sub>8</sub> transition. Ho lasers find applications in coherent laser radar, atmospheric sensing, surgery and frequency conversion into mid-IR. Among the laser host crystals for Ho<sup>3+</sup> doping, garnets (Y<sub>3</sub>Al<sub>5</sub>O<sub>12</sub>) and sesquioxides (A<sub>2</sub>O<sub>3</sub>, A = Lu, Y, Sc) have attracted attention due to their attractive thermal and spectroscopic properties. Transparent polycrystalline ceramics offer the advantages of lower synthesis temperatures, ease of doping and size-scalable production [1]. We report on a comparative study of Ho:Y<sub>3</sub>Al<sub>5</sub>O<sub>12</sub> and Ho:Y<sub>2</sub>O<sub>3</sub> transparent ceramics promising for thin-disk ~2 μm lasers.

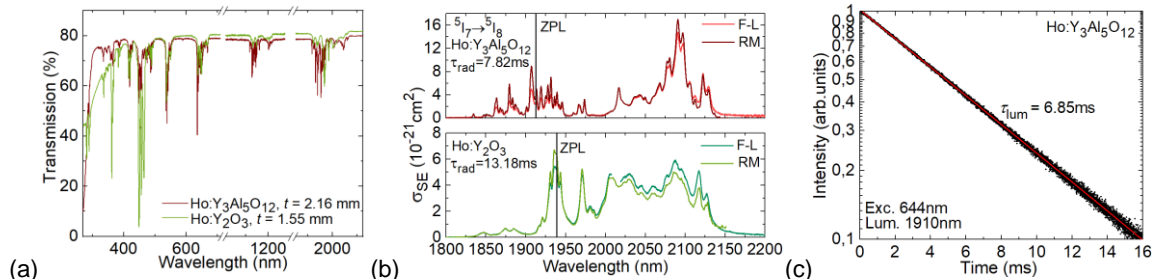
The 0.94 at.% Ho:Y<sub>3</sub>Al<sub>5</sub>O<sub>12</sub> ceramic was fabricated by vacuum sintering at 1780 °C for 20 h using 0.5 wt% TEOS as a sintering aid from Ho:Y<sub>2</sub>O<sub>3</sub> and Al<sub>2</sub>O<sub>3</sub> nanopowders produced by laser ablation and sedimentation [2]. For the 0.94 at.% Ho:Y<sub>2</sub>O<sub>3</sub> ceramics, we used laser-ablated Ho:Y<sub>2</sub>O<sub>3</sub> nanopowder with an addition of ~2 wt% ZrO<sub>2</sub> as a sintering aid to avoid exaggerated grain growth. They were uniaxially pressed and vacuum sintered at 1780 °C for 20 h. The obtained ceramics were transparent with yellowish coloration, Fig. 1(a).

The ceramics were of single-phase nature. The Ho:Y<sub>3</sub>Al<sub>5</sub>O<sub>12</sub> one exhibited a cubic structure (sp. gr. *Ia*3<sup>-</sup>*d*, *a* = 10.010 Å), Fig. 2(a), and a mean grain size *D*<sub>grain</sub> of 32 μm. The Ho:Y<sub>2</sub>O<sub>3</sub> ceramics was also cubic (sp. gr. *Ia*3<sup>-</sup>, bixbyite (C-type) structure, *a* = 10.569 Å) with smaller *D*<sub>grain</sub> of 6.8 μm. Both ceramics exhibited a close-packed microstructure with clean grain boundaries and a lack of pores, Fig. 1(c).



**Fig. 1** (a) Photograph of the laser-grade polished ceramic disks; (b) X-ray powder diffraction (XRD) patterns of ceramics; (c) Scanning Electron Microscope (SEM) image of the fracture surface of the Ho:Y<sub>2</sub>O<sub>3</sub> ceramics.

The ceramics exhibited high transmission in the visible and IR, Fig. 2(a). For the Ho:Y<sub>2</sub>O<sub>3</sub> ceramics at 1.0 μm, *T*<sub>0</sub> = 80.0% which is close to the theoretical limit set by Fresnel losses, *T*<sub>Fr</sub> = 81.6%. The transition intensities were analysed within the Judd-Ofelt theory. Absorption,  $\sigma_{\text{abs}}$ , and stimulated-emission (SE),  $\sigma_{\text{SE}}$ , cross-sections, Fig. 2(b), for the <sup>5</sup>I<sub>7</sub> ↔ <sup>5</sup>I<sub>8</sub> Ho<sup>3+</sup> transition were determined and the luminescence lifetimes of several Ho<sup>3+</sup> excited-states were measured. The multi-phonon non-radiative (NR) relaxation rates were estimated. For the Ho:Y<sub>3</sub>Al<sub>5</sub>O<sub>12</sub> ceramics,  $\tau_{\text{lum}}(^5\text{I}_7) = 6.85$  ms, Fig. 2(c), which is close to the estimated radiative lifetime (7.82 ms).



**Fig. 2** Spectroscopy of Ho:Y<sub>3</sub>Al<sub>5</sub>O<sub>12</sub> and Ho:Y<sub>2</sub>O<sub>3</sub> transparent ceramics: (a) in-line transmission spectra; (b) stimulated-emission,  $\sigma_{\text{SE}}$ , cross-sections for the <sup>5</sup>I<sub>7</sub> → <sup>5</sup>I<sub>8</sub> transition calculated by the Fuchtbauer-Ladenburg (F-L) equation and reciprocity method (RM); (c) luminescence decay curve from the <sup>5</sup>I<sub>7</sub> Ho<sup>3+</sup> state for the Ho:Y<sub>3</sub>Al<sub>5</sub>O<sub>12</sub> ceramics.

### References

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